# EFFECT OF Li<sub>5</sub>AlO<sub>4</sub> ADDITIONS ON THE SINTERING AND PROPERTIES OF MAGNESIUM ALUMINATE SPINEL

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#### Abstract

Magnesium aluminate spinel (MgAl<sub>2</sub>O<sub>4</sub>) with high transparency has been wildly used in various applications because of its high mechanical strength and good optical properties. The aim of this work is to evaluate the effects of an additive, Li<sub>5</sub>AlO<sub>4</sub>, on the sinterability of MgAl<sub>2</sub>O<sub>4</sub>. The concentrations of Li<sub>5</sub>AlO<sub>4</sub> focused in this study were 1, 5, and 10 wt%. An aqueous solution of Li<sub>5</sub>AlO<sub>4</sub> was prepared by the reaction between high purity lithium nitrate and aluminium nitrate. Under the magnetic stirring at  $60^{\circ}$ C, MgAl<sub>2</sub>O<sub>4</sub> powder (Baikalox, S30CR) was homogeneously added into the solution. The obtained viscous slurries were then dried, ground, sieved, and uniaxially pressed under 50 MPa. The green bodies were sintered in air at atemperature of  $1600^{\circ}$ C. It was found that, for the 1 wt% Li<sub>5</sub>AlO<sub>4</sub> added specimens sintered at  $1600^{\circ}$ C, the relative density reached up to 99.47%. The pre-sintered specimens were subjected tohotisostatic pressing at  $1500^{\circ}$ C for 1 h. Phase compositions, microstructures, and bulk densities of the sintered and HIP specimens were also characterized and discussed.

Keywords: Magnesium aluminate spinel, Li<sub>5</sub>AlO<sub>4</sub>, hot isostaticpressing, sintering

#### Introduction

Nowadays, transparent ceramics are widely used in many applications. They are used as window glass and decoration for the construction industries and are used as lenses for specific work whenoptical properties are necessary. Including military usage as an application for transparent armour, transparent ceramics have been of interestwhen there is a requirement for excellent optical properties,

high chemical stability, and good mechanical strength.

Many kinds of transparent ceramics, such as aluminum oxynitride (AlON) (Clay et al., 2006; Frage et al., 2007), alumina, and yittria alumina garnet (YAG) (Li et al., 2012) have been widelydeveloped and used in various engineering fields. One of the most suitable materials for transparent ceramics is magnesium

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aluminate spinel (MgAl<sub>2</sub>O<sub>4</sub> spinel) because of its high mechanical strength, good optical properties, and beinglighter than metal (Goldstein et al., 2008). MgAl<sub>2</sub>O<sub>4</sub> spinel has excellent optical properties with useful transmission from about 0.2 nm in the UV through the visible range to about 6.0 nm in the near-IR, while its cubic structure allows it to be made transparent, even at a considerable thickness, while maintaining decent mechanical properties (Dericioglu and Kagawa, 2003; Krell et al., 2010). MgAl<sub>2</sub>O<sub>4</sub> spinel and sapphire are bothcandidates for transparent armor systems. MgAl<sub>2</sub>O<sub>4</sub> spinel has the advantage of easier processing and a lower cost than sapphire.

Relatively expensive powders and high finishing costs prevent the wide-spread use of sapphire. The present research investigated the use of commercial spinel powder as the primary component in the starting powders and hypothesized thata sintering additive, such as lithium aluminate (Li<sub>5</sub>AlO<sub>4</sub>) with a melting point of 1055°C would aid in densification via transient liquid phase sintering. A beneficial side effect of using lithium aluminate as a transient liquid phase is that it can be removed from the sintered specimens by evaporation at a higher temperature (over 1300°C) due to its high vapor pressure. The aim of this work is to evaluate the effect of Li<sub>5</sub>AlO<sub>4</sub> as a sintering additive on the sinterability of MgAl<sub>2</sub>O<sub>4</sub> spinel.

# **Materials and Methods**

## **Preparation of Specimens**

Various fabrication technologies were developed (Goldstein *et al.*, 2009). In this paper we used 2 stages of heat treatment, air sintering and hot isostatic pressing (HIP) (Krell *et al.*, 2010; Benameur *et al.*, 2011). An aqueous solution of Li<sub>5</sub>AlO<sub>4</sub> was prepared by a reaction between high purity lithium nitrate (Fluka 98.0%, Sigma-Aldrich Corp., St. Louis, MO, USA) and aluminium nitrate nanohydrate (QRËCTM grade AR) that

wereused as starting materials for the sintering additives. An aqueous solution of nitrate precursors was prepared by magnetic stirring. The concentrations of Li<sub>5</sub>AlO<sub>4</sub> to the spinel powder in this study were 1, 5, and 10 wt%. Commercial MgAl<sub>2</sub>O<sub>4</sub> spinel powder (SCR30, 99.0%, Baikarox) was homogeneously added into the prepared nitrate solution. The viscous slurries that were obtained had been heated andstirred until dried at about 100°C. The mixtures were then dried again at 100°C in an oven, ground, and sieved through a 100 mesh. The sieved powders were uniaxially pressed at 50 MPa into pellet shapes with a 2.5 mm in diameter. The pellet green bodies were sintered in an electrical box furnace at the temperature range 1100-1600°C for 2 h. After the air sintering stage, HIP was performed at 1500°C/1 h (Ar, 200 MPa). The flow chart of the preparation is shown in Figure 1.

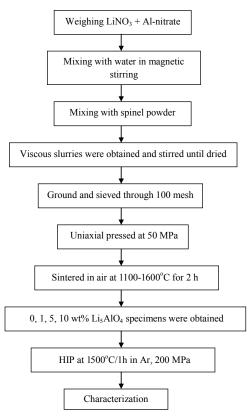


Figure 1. The flow chart of preparations

#### Characterization

The bulk density and water absorption were measured by the standard method according to the American Society of Testing and Materials (ASTM C373-88 (2006)). Shrinkage and weight loss were measuredby length and weight before and after firing. The crystalline phases were investigated by X-ray diffraction (XRD) (D8-Advance, Bruker Corp., Billerica, MA, USA)using Cu-Ka radiation, and the diffraction data were collected over the 20 range from 5 to 60° with a step size of 0.02°. The microstructures were investigated by optical microscope (Olympus BX60M, Olympus Corp., Tokyo, Japan) and scanning electron microscopy (SEM)(JSM 6480 LV, JEOL Ltd., Tokyo, Japan).

#### **Results and Discussion**

The firing shrinkage of 4 different concentrations of Li<sub>5</sub>AlO<sub>4</sub> is shown in Figure 2. The firing shrinkage in percentage increased with the increased temperature. The sample without the additive has higher shrinkage than that of the other concentrations.

The water absorption of the specimens after firing from 1100 to 1600°C is shown in Figure 3. It indicated that water absorption was less than 0.03% in the 1 wt% added specimen.

In addition, Figure 4 presents the percentage of weight loss of the MgAl<sub>2</sub>O<sub>4</sub> spinel specimens after sintering which are dependent on the temperature and concentration of Li<sub>5</sub>AlO<sub>4</sub> added.

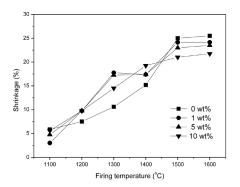


Figure 2. Effect of  ${\rm Li}_5{\rm AlO}_4$  additions on firing shrinkage of  ${\rm MgAl}_2{\rm O}_4$  spinel

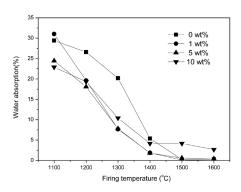


Figure 3. Effect of  $\mathrm{Li}_5\mathrm{AlO}_4$  additions on water absorption of specimens

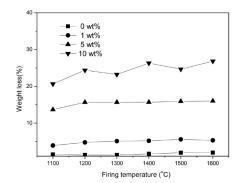


Figure 4. Effect of Li<sub>5</sub>AlO<sub>4</sub> additions on weight loss

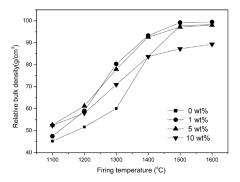


Figure 5. Effect of Li<sub>5</sub>AlO<sub>4</sub> additions on bulk density

The bulk density of the samples is shown in Figure 5. The bulk density increased when the temperature increased. The highest bulk density was found in the 1 wt% added, which reached up to 99.47% when sintered

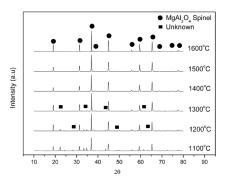


Figure 6. XRD patterns of 10 wt% Li<sub>5</sub>AlO<sub>4</sub> specimens sintered at temperature range of 1100-1600°C

at 1600°C. However, all of the specimens were still not transparent.

The XRD patterns of the crystalline phases of the specimens are shown in Figure 6. In the temperature range between 1100-1300°C, there are 2 phases, MgAl<sub>2</sub>O<sub>4</sub> spinel and an unknown phase found as a secondary phase. At the higher temperature range of 1400-1600°C, the secondary phase has disappeared due to the higher sintering temperature. Only 1 phase, MgAl<sub>2</sub>O<sub>4</sub> spinel, remained. According to Figure 5, using Li<sub>5</sub>AlO<sub>4</sub> as the sintering additive can increase the bulk density and it can be eliminated at over 1400°C.

Figure 7, from left to right, presents the microstructure of the specimens sintered at 1600°C (a) without additive, (b) 1 wt%, (c) 5 wt%, and (d) 10 wt% of the Li<sub>5</sub>AlO<sub>4</sub> additions, respectively. At 1600°C, a suitable

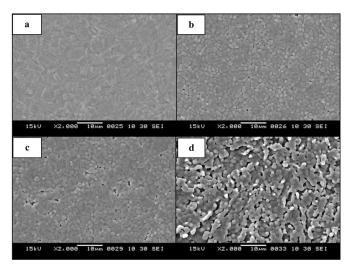


Figure 7. SEM micrographs of spinel specimens after sintering at  $1600^{\circ}\text{C}$ ; (a) without additive, (b) 1 wt%, (c) 5 wt%, (d) 10 wt% of Li<sub>5</sub>AlO<sub>4</sub>

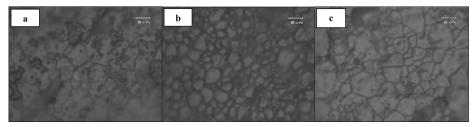


Figure 8. Optical micrographs of spinel specimens after sintering at 1600°C; (a) without additive, (b) 1 wt%, (c) 5 wt%

concentration of Li<sub>5</sub>AlO<sub>4</sub> was 1 wt% that not only decreased the grain size, but also increased the densification. A 1 wt% addition was related to the result of the water absorption in Figure 3. And in Figure 7(d) the highestaddition made an increase in porosity because of the evaporation of the lithium compound.

## **Conclusions**

In summary, the effect of the  $\text{Li}_5\text{AlO}_4$  additions on the sintering of MgAl<sub>2</sub>O<sub>4</sub> spinel showed that a small grain size, low firing shrinkage, and high densification at  $1600^{\circ}\text{C}$  were obtained by using 1 wt% of  $\text{Li}_5\text{AlO}_4$  addition. On the other hand, weight loss was increased.

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